



Development of Cupric Oxide/Porous Silicon (CuO/PS) Heterostructure Enabled Room Temperature Methane Sensor for Enhanced Industrial Safety

Khaniyev Bakyt,¹ Ibraimov Margulan,¹ Nalibayev Yerkebulan,¹ Skabylov Alisher,¹ Khaniyeva Ainur,¹ Jianxi Liu,² Tezekbay Yerbolat,¹ Duisebayev Tolagay,¹ Tileu Ayan¹ and Meirambekuly Nursultan^{1,*}

Abstract

We report the development and characterization of a methane gas sensor made from copper oxide/porous silicon (CuO/PS) heterostructure. The sensor integrates the CuO layer on the PS surface, obtained by the DC magnetron sputtering technology, which increases the sensor's surface area and facilitates gas molecule interaction. We present an analysis of the sensor's morphological and structural characteristics, including detailed scanning electron microscopy (SEM) and X-ray diffraction (XRD) studies, and revealed the effective formation and interface of the heterostructure. The sensor demonstrates performance with good sensitivity to methane, characterized by a dynamic response. Sensitivity tests show a significant enhancement in signal response, and dynamic response evaluations indicate the repeatability of the sensor. The results suggest that the CuO/PS sensor offers a promising solution for methane detection, with potential applications in environmental monitoring, industrial safety, and process control.

Keywords: Methane gas sensor; Semiconductor heterostructure; Copper oxide; Porous silicon; Monitoring system; DC magnetron sputtering.

Received: 30 July 2024; Revised: 17 September 2024; Accepted: 18 September 2024.

Article type: Research article.

1. Introduction

Methane is a colorless and odorless gas that is used as fuel and raw materials in various industrial sectors.^[1] However, methane leaks in large quantities pose serious risks.^[2] Methane is highly flammable and can form explosive mixtures with air at concentrations from 4.9% to 15%. This makes methane leaks dangerous and potentially destructive.^[3-7] In addition, methane leaks can pollute the air, soil and reservoirs, which can also have negative consequences for human health and ecosystems.^[8,9] To minimize the risks associated with methane or other types of gases, it is extremely important to have effective systems for its detection.^[10,11] Methane monitoring technologies and sensors, which can detect gas in low concentrations in real time, play a key role in ensuring safety

at industrial facilities. These systems allow timely detection of leaks and prevention of their consequences, which helps to protect human health and maintain environmental balance.^[11] The integration of reliable methane monitoring and detection systems is an essential component of industrial safety management. These systems help to prevent potential accidents, protect people and the environment, and ensure the safe operation of industrial facilities.^[12-14]

Many studies have been conducted on methane detection, in which various approaches have been proposed, such as photonic crystal micro-cavity, hollow-core photonic band-gap fiber, surface plasmon resonance, utilization of the Vernier effect in fiber optic interferometers and spectroscopic methods.^[15-18] Among all sensor technologies, optical and spectroscopic sensors are widely investigated.^[19,20] Optical sensors for the detection of methane typically work by analysing the absorption of light at specific wavelengths that correspond to the absorption characteristics of methane. Zhang *et al.* developed an innovative optical methane sensor that integrates cryptophane-E into a photonic-crystal micro-cavity, combining the selective adsorption capabilities of cryptophane-E for methane with the high resonance

¹ Department of Physics and Technology, Al-Farabi Kazakh National University, Al-Farabi av. 71, Almaty 050040, Kazakhstan.

² State Key Laboratory of Solidification Processing, Center of Advanced Lubrication and Seal Materials, School of Materials Science and Engineering, Northwestern Polytechnical University, Xi'an 710072, P.R. China.

*Email: nurs.kaznu@gmail.com (M. Nursultan)

characteristics of the photonic-crystal micro-cavity.^[15] Martyniuk et al. reported on the Ga-free InAs/InAsSb type II superlattice (T2SL) based long wavelength infrared (LWIR) methane sensor operating at $\lambda \sim 8 \mu\text{m}$ and capable of detecting methane at the ppb level in the LWIR region.^[21] Yang *et al.* presented a photonic-crystal fiber methane sensor based on modal interference with a good sensitivity and a low detection limit.^[22] However, the potential for widespread use of such sensors in practice is limited by the high cost and structural complexity of optical sensors, their high sensitivity to environmental conditions such as temperature and humidity, and their susceptibility to dust and other small particles that can affect the accuracy of methane detection. Therefore, it is necessary to find other methods or technologies for methane detection. One of these methods is the use of gas sensors based on metal oxide semiconductors (MOS) to detect methane.^[23] MOS based gas sensors have several advantages that make them popular for various applications including methane detecting, as low cost, small size, high sensitivity, fast response time, and simplicity of manufacturing.^[24-26] Many studies have been conducted to investigate the characteristics of methane sensors based on MOS, such as SnO_2 , ZnO , TiO_2 , Co_3O_4 , and WO_3 .^[27-32]

One of the MOS materials with unique optical, electrical, and morphological properties that has a high potential for use in gas sensor technology is CuO .^[33] CuO band gap is 1.2 eV, and it is characterized by p-type conductivity.^[34] Kim *et al.* created interconnected p- CuO nanowires through thermal oxidation which can detect 10 ppm of NO_2 at an operating temperature in the range of 200 °C-400 °C.^[35] Lai *et al.* fabricated a gas sensor using CuO nanowires on patterned interdigitated electrodes, which can detect ozone gas at a low concentration of 50 ppb at 100 °C.^[36] The main drawbacks of MOS sensors include operating at high temperatures and poor compatibility with modern silicon-based electronic devices. In order to increase the sensitivity and reduce the operating temperature, CuO films with a porous structure and high surface activity can be prepared. Thus, through the deposition of CuO on a porous silicon (PS) surface and the formation of

a heterostructure structure, it is possible to improve its sensing performance at room temperature. PS combines high specific surface area, chemical inertness, ease of integration with microelectronic systems and thermal stability, making it an ideal choice as a substrate for gas sensors. These properties ensure high sensitivity, reliability and durability of sensors, which is critical for their effective operation in various applications.^[37,38] There have been some reports on the gas detection of PS/ WO_3 , PS/ ZnO , PS/ V_2O_5 , PS/ TeO_2 composites.^[39-43] Nevertheless, the methane sensing properties of PS/ CuO heterostructures have not been reported.

In this paper, the CuO/PS heterojunction structure and its optical, morphological, and sensing properties for methane detection are presented.

2. Experimental section

In this work the PS samples are used as substrates of the heterostructure material. The PS samples were produced through the electrochemical etching method, a widely used technique due to its effectiveness and relative simplicity. A porous layer was formed on the surface of a p-type silicon wafer, with an orientation and resistance of 10 $\Omega\cdot\text{cm}$. To facilitate handling, the silicon wafer was cut into $1 \times 1 \text{ cm}^2$ pieces (Fig. 1a). Before initiating the etching process, the silicon wafer was immersed in a hydrofluoric acid (HF) solution for 10 seconds to remove any impurities and then cleaned with ethanol to ensure a clean surface. The electrochemical etching was performed in a fluoroplastic Teflon cell using an electrolyte with a 1:1 volume ratio of HF and ethanol. The etching process was performed for 40 min with a stable current density of 5 mA/cm^2 . The samples were then rinsed with deionized water and left to dry in the air. The porosity was determined using the gravimetric method (Fig. 1b).

The copper oxide (CuO) layer in the PS surface were prepared through magnetron sputtering followed by annealing. The magnetron sputtering process was carried out using the Kurt J. Lesker LAB-18 magnetron system. (Kurt J. Lesker Company, Dresden, Germany). A CuO target with a purity of

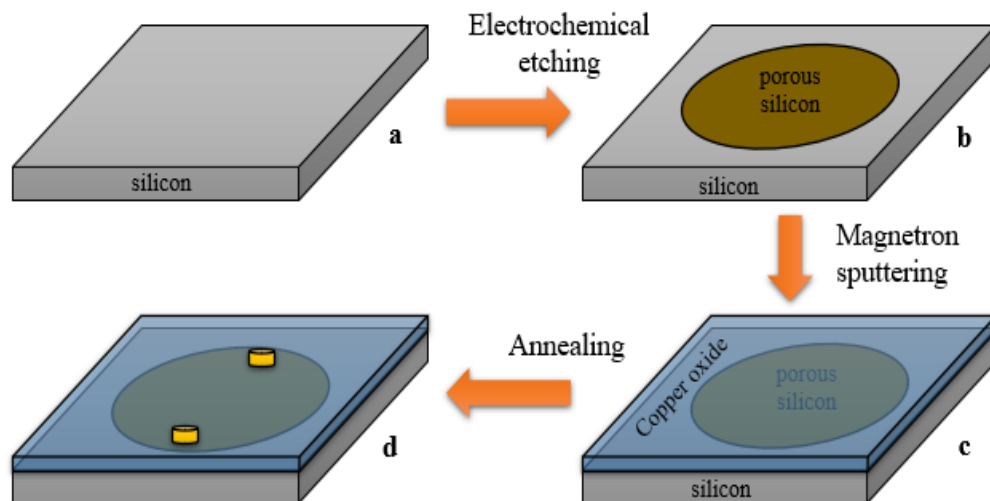


Fig. 1 Schematic illustration of the process for forming CuO/PS gas sensor.

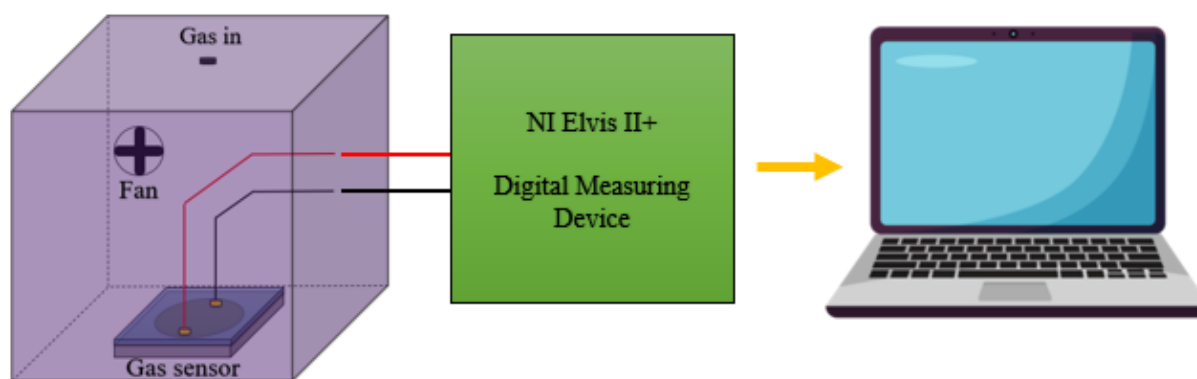


Fig. 2 Schematic diagram of the gas sensing measurement system.

99.999% was used to deposit the CuO layer in an atmosphere of argon (Ar) and oxygen (O₂). Ar and O₂ gases were introduced into the chamber in a ratio of 4:1. Deposition of the CuO layer on the PS samples was carried out for 30 min at power 100W, gas pressure 10.5 mTorr, and substrate-to-target distance of 13 cm. After the formation of CuO on the surface of the PS sample, the sample was kept in a furnace at a temperature of 650 °C for 4 h to crystallize (Fig. 1c).

The thickness of the CuO layer on the PS surface was measured using a profilometer Dektak XT Stylus profilometer (Bruker, Billerica, Massachusetts, USA). The morphology of the samples was analyzed using a scanning electron microscope JSM-IT2000 (JEOL Ltd., Tokyo, Japan). The XRD spectra, which were used to examine the phase formation of the samples, were acquired using a Rigaku Miniflex 600 diffractometer (Rigaku, Tokyo, Japan). Structural analysis was done by Raman spectroscopy using NT-MDT Solver Spectrum with a laser excitation wavelength of 473 nm.

Two ohmic contacts of InGa alloy were deposited on the surface of the samples in a coplanar configuration using thermal deposition to measure their electrical characteristics (Fig. 1d). The gas sensing measurement was carried out at room temperature using a chamber with a volume of 8L (Fig.

2). The base gas in the measurement is pure air. The current-voltage characteristics and sensing response curves were measured with NI Elvis II+ module.

3. Results and discussion

The morphology of the PS and the CuO/PS samples were observed from the scanning electron microscopy (SEM) images, which are shown in Fig. 3. The SEM image of the PS sample reveals a highly textured surface characterized by numerous interconnected pore-like structures (Fig. 3a). The surface is predominantly covered with an intricate network of pores, varying in size and shape. These pores are distributed uniformly across the surface, giving the material a sponge-like appearance. The porosity value was 72%. The cross-section view of the PS shows aligned channels. Fig. 3b shows the surface of the CuO layer deposited on PS, revealing a uniform porous structure, so high porosity, large specific surface area as well as diffusion conduits for gas molecules can be achieved. The thickness of the CuO layer was measured by using a profiler and amounted to 132 nm.

The structure of the CuO/PS sample was characterized using XRD and Raman spectroscopy (Fig. 4). The Raman spectrum of the CuO/PS sample shows typical characteristic

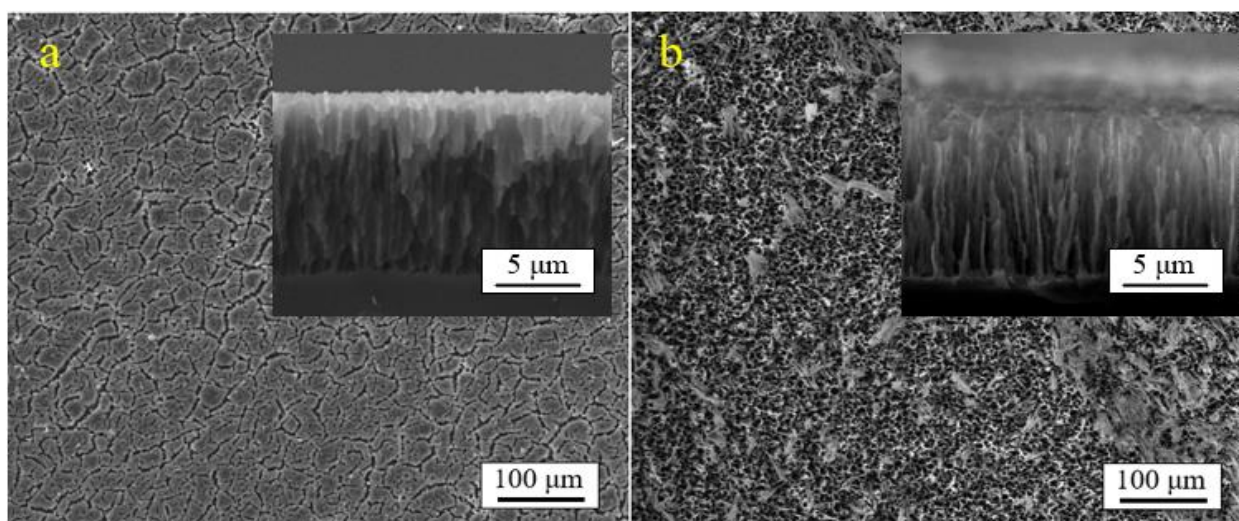


Fig. 3 Top-view and cross-sectional SEM images of the PS (a) and the CuO/PS (b) samples.

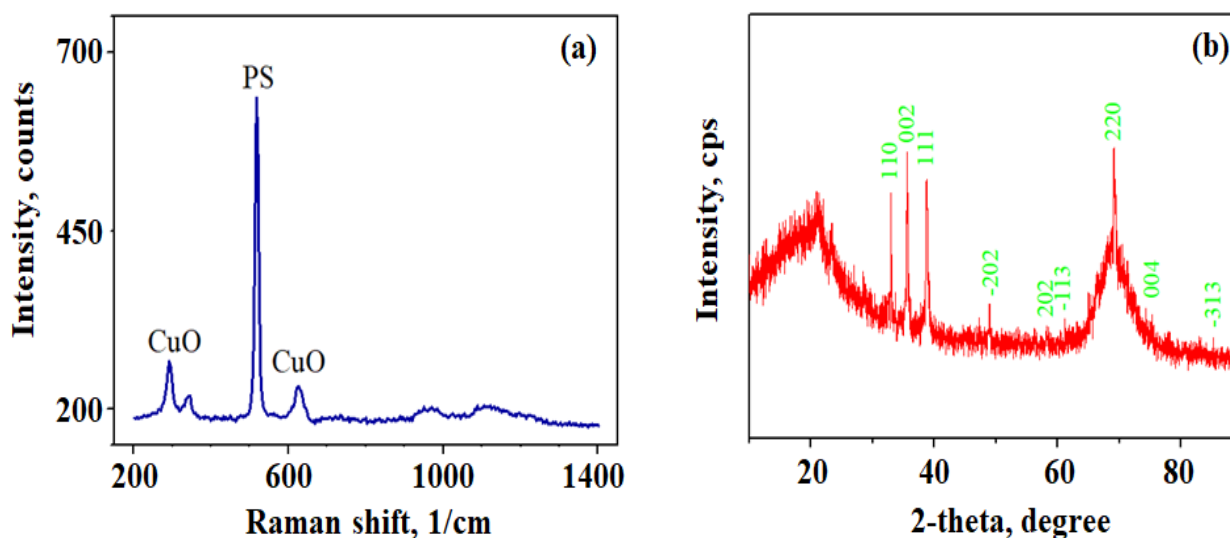


Fig. 4 Raman spectrum (a) and XRD pattern (b) of the CuO/PS sample.

peaks associated with both CuO and the PS substrate (Fig. 4a). For CuO, prominent peaks corresponding to the bending and stretching vibrations of the Cu–O bonds are observed, particularly around 300 cm⁻¹ and 620 cm⁻¹, indicative of the presence of CuO in its different phases.^[44] In addition, the PS component contributes a peak around 520 cm⁻¹, associated with the Si–Si bond vibrations.^[45]

The XRD pattern of the CuO/PS heterostructure reveals distinct diffraction peaks corresponding to the crystalline phases of CuO and PS. Peaks typically observed for CuO include those at 2θ values of approximately 35.6°, 38.7°, 48.8°, and 61.5°, corresponding to the (110), (111), (202), and (220) planes of the monoclinic CuO phase, respectively. A sharp diffraction peak can be seen at 69.2° for the as-prepared PS sample, corresponding to the plane reflection of Si.^[46] Additionally, there is a broad hump in the region of 15–30° and it is likely related to the formation of an amorphous SiO₂ phase.^[47] The strong diffraction peaks of CuO appear at 35.662°, 38.853°, and 48.954°, corresponding to the (110), (111), and (202) planes of the monoclinic CuO phase, respectively.^[48] These peaks confirm the presence and crystalline nature of CuO within the heterostructure. The values match precisely with the standard data (PDF Card No.: 00-041-0254).

Figure 5 depicts the current-voltage characteristics of the CuO/PS sample measured under the influence of 10 ppm methane and in air at room temperature. In air, the current-voltage curve exhibits a baseline behavior where the current changes predictably with applied voltage, showing a quiet linear relationship. When exposed to methane with a concentration of 10 ppm, the current-voltage characteristics changed, reflecting the sample's response to the gas. One might observe an increase in current due to the interaction between methane molecules and the sample's surface, which can alter its electrical conductivity. This increase in current may be due to enhanced charge carrier activity or changes in the sample's electronic structure caused by the gas interaction.

It was determined that the sensor current reaches saturation at a certain concentration of methane gas. Therefore, the CuO/PS gas sensor has an upper limit of gas concentration at which the current value is saturated.

Figure 6 shows the dynamic response curve of the CuO/PS sample when exposed to methane at a concentration of 10 ppm at room temperature. It can be seen that the current increases dramatically upon exposure to methane and recovers after clean air flows into the chamber. The response/recovery time of the sensor can be determined from the dynamic response of the sensor and amounted to 20 sec and 100 sec, respectively, which are similar with other researches.

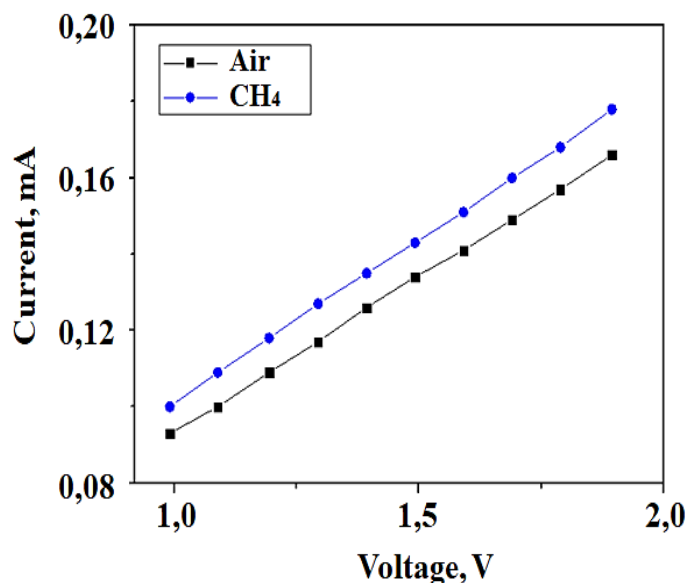


Fig. 5 The current-voltage characteristics of the CuO/PS sample under the influence of 10 ppm methane and in air.

Measurements showed that the sensor response of the CuO/PS sensor to methane gas reaches 7%. The sensor response of the sensor was calculated as:^[49]

$$S_R = \frac{I_{gas} - I_{air}}{I_{air}} \quad (1)$$

where I_{air} and I_{gas} are the currents (resistances) measured in the air and target gas (methane) ambiences, respectively.

An energy band diagram can be used to explain the sensing mechanism of the CuO/PS sample. The band gap of PS is larger than that of silicon due to the quantum confinement effect.^[50] The electronic affinity of PS is similar with Si (4.01 eV). As for CuO, also a p-type semiconductor, E_g (CuO) = 1.35 eV, χ (CuO) = 4.07 eV.^[51] When these two semiconductors come into contact with each other, a p-n heterostructure is formed at the interface. Due to the difference in Fermi energies, electrons from CuO with higher energies flow from CuO to PS, while holes flow in the opposite direction until the Fermi levels become equal. This results in a depletion layer on the PS side and a hole accumulation layer on the CuO side. Due to the different dielectric constants of CuO and PS, the energy bands become discontinuous. This efficient charge separation results from the heterostructure, which offers a high concentration of charge carriers in the accumulation layer, increasing the lifetime of these carriers. Consequently, the material facilitates easier exchange of electrons with adsorbed gases.

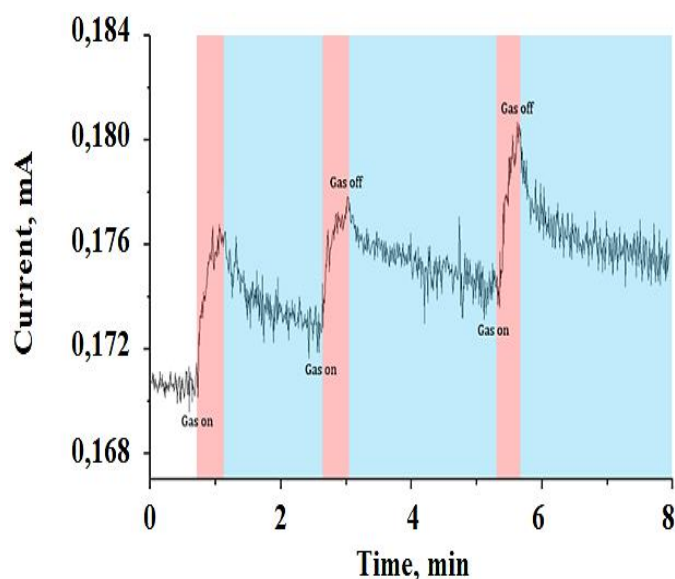


Fig. 6 The dynamic response of the CuO/PS sample to methane.

When the sensor is exposed to air, oxygen molecules are adsorbed onto its surface. Due to the sensor's large specific surface area, it offers numerous active sites for gas adsorption. This extensive surface area increases the number of active centers available for oxygen molecule attachment. Consequently, a larger specific surface area results in a higher density of oxygen vacancies on the material's surface. As a result, the sensor can adsorb a significant number of oxygen molecules even at room temperature. Oxygen molecules adsorbed on the surface exchange with electrons on the sensor surface to form oxygen ions O_2^- which are stable at the working room temperature. As a result, the energy band of the CuO/PS heterostructure becomes further bent. When gas molecules interact with the sensor, an acidification reaction occurs between the gas molecules and oxygen anions. This

reaction increases the concentration of defects in the accumulation layer and reduces the sensor's resistance.

4. Conclusions

In conclusion, this work demonstrates the promising potential of the methane gas sensor prepared using CuO/PS heterostructure. Through detailed analysis of the morphological and structural characteristics, the heterostructure's unique interface significantly enhances the sensor's functional properties was established. The intricate morphology of the PS and the well-defined structure of the CuO layer contribute to the effective interaction with methane gas molecules, leading to improved sensitivity. The sensor exhibits notable sensitivity (7%) and dynamic response to methane at room temperature. The gas sensing mechanism was explained using the energy band diagram and was attributed to the synergistic effects of CuO and PS materials, which facilitate efficient charge transfer and enhance gas adsorption.

Our findings suggest that this sensor architecture not only provides a reliable and efficient means for methane detection but also paves the way for further advancements in gas sensing technology. Future research will focus on further optimization of the sensor's material composition and structure, as well as exploring its long-term stability, selectivity and performance in diverse environments. The insights gained from this study provide a solid foundation for developing advanced methane sensors with improved accuracy and reliability, with potential implications for environmental monitoring, safety, and industrial applications.

Acknowledgements

This research was funded by the Science Committee of the Ministry of Science and Higher Education (MHES) of the Republic of Kazakhstan under Grant No. AP19678266.

Conflict of Interest

There is no conflict of interest.

Supporting Information

Not applicable.

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